

HIP

Atomistic approach in simulations of electrical breakdowns on metal surfaces

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Behind the model...

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HIP







Doc. Flyura Djurabekova Senior scientist Prof. Kai Nordlund M Sc Aarne Pohjonen Dislocations



M Sc Stefan Parviainen M Field emisssion and C neutral atom evaporation

nen MSc Avaz Ruzibaev d Charges on surfaces



Dr Juha Samela Sputtering and cratering



M Sc Helga Timko Plasma simulations (CERN, Switzerland)



Dr Lotta Mether Plasma simulations CERN, Switzerland



Dr. Walter Wuensch

Sergio Calatroni





 Multiscale model to approach the problem of electrical breakdown
 Surface charge, workfunctions
 Dislocations as a media of surface response to electric fields

Electric discharges near a metal surface

Summary







Multiscale model to simulate electrical breakdown



Evolution of a tip placed on Cu surface



Details in F. Djurabekova, S. Parviainen, A. Pohjonen and K. Nordlund, PRE 83, 026704 (2011).

Solution We developed a novel approach to follow the dynamic evolution of partial charge on surface atoms by combining the MD and classical ED (solving Laplace equation)

- The dynamics of atom charges follows the shape of electric field distortion on tips on the surface
- Temperature on the surface tips is sufficient => atom evaporation enhanced by the field can supply neutrals to build up the plasma densities above surface.



DFT calculations to validate the charges on surface atoms





s DFT details:

Code: SIESTA

- For exchange and correlations functionals the Perdew, Burke and Ernzerhof scheme of Generalized gradient approximation (GGA)
- Slab organized in 8 layers + 8 layers of vacuum
- External field is added to calculate the electrostatic potential in the vacuum

$$\sigma = \varepsilon_0 \vec{E} = 5.53 \times 10^{16} \, \frac{\bar{e}}{m^2} \Leftrightarrow \sigma = \frac{Q_{surf}}{A_{surf}} = 5.49 \times 10^{16} \, \frac{\bar{e}}{m^2}$$

| | An adatom | | Double adatom | |
|--|-------------|---------|---------------|---------|
| | DFT, SIESTA | ED&MD | DFT, SIESTA | ED&MD |
| Charge (q _e) per adatom | -0.032 | -0.0215 | -0.025 | -0.0177 |



Motivations: why we look for dislocations?

The dislocation motion is strongly bound to the atomic structure of metals. In FCC (face-centered cubic) the dislocation are the most mobile and HCP (hexagonal close-packed) are the hardest for dislocation mobility.





Experimental evidence of the voids

SEM images by Dr. Tomoko Muranaka (Uppsala Univ.)





Voids: a lattice irregularity which can be a source of a protrusion growth



Solution We simulated a void near {110} Cu surface, when the high tensile stress is applied on the surface. Bottom is fixed, lateral boundary allowed to move in z direction.

 $\uparrow \uparrow \uparrow \uparrow \uparrow \uparrow \uparrow \uparrow$



A. Pohjonen, F. Djurabekova, et al., Dislocation nucleation from near surface void under static tensile stress on surface in Cu, *Jour. Appl. Phys.* 110, 023509 (2011).

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Concurrent ED-MD simulations of dislocations on a near-surface void

Shalf-void of diameter 4nm in {110} Cu surface. (N of atoms≈ 170000 atoms...)

so $E_0 = 22 \text{ GV/m}$ (exaggeration is required to simulate the dislocation within the MD time span)

∽ T = 600 K







http://indico.cern.ch/conferenceDisplay.py?confld=8831.] with the model.] Flyura Djurabekova, HIP, University of Helsinki



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From tips to plasma: From FE to discharge currents

so In real life we can observe the full dynamic range of a vacuum discharge:

- > > 10s pA in 'weak' FE phase
- \succ Space charge limited 'strong' FE phase, typically ~ nA μA
- > Discharge current, up to 10 100 A

At the same time, the involved area changes:

- > Typically $10^{-20} 10^{-14} \text{ m}^2$ for weak FE \Rightarrow R_{em} ~ 0.1 100 nm
- \succ During the discharge, the bombarded area has R \sim 10 100





Up to 12 orders of magnitude difference

Up to 12 orders of magnitude difference



Symmetry Solution of the symmetry

- \triangleright Particles: e⁻, Cu, and Cu⁺
- Monte Carlo collision routines (Max Plank Institute of Plasma Physics, K.Matyash)
- s Emission processes



- > Fowler-Nordheim field emission enhanced by β
- Simplified Cu evaporation fraction of FN emission
- > Sputtering (*Yamamura & Tawara*)
- Heat spike sputtering from MD simulations
- > Secondary electron yield constant
- > lons only through impact: $e^- + Cu \rightarrow 2e^- + Cu^+$
- s External RC circuit
 - > Potential stored in capacitor
 - Drained by arc current

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high β (initially)

Rest of surface, low β (constant)



Plasma build-up

Two conditions need to be fulfilled:

- High enough <u>initial local field</u> to have growing FE current
- Reaching the <u>critical neutral density</u> to induce an ionisation avalanche



H. Timko, et al. A One-Dimensional Particle-in-Cell Model of Plasma Buildup in Vacuum Arcs, Contrib. Plasma Physics 51, 5 (2011).

From field emission to developed arc - a typical 2D Arc-PIC simulation

Field emission + ionization
 Sudden ionization avalanche
 Sheath + quasi-neutral plasma forms
 Plasma self-maintaining if energy is available
 Neutrals fill entire gap

 $\begin{array}{ll} \beta_0 = 35, \, \beta_f = 2 \;\; Grid \; 240 \times 400 \\ E_{ext} = 290 \; MV/m \;\; C_{ext} = 1 \; nF \\ r_{Cu/e} = 0.015 \;\; R_{em} = 0.4 \; \mu m \end{array}$

H. Timko, L. Mether et al., Modeling of cathode plasma initiation in copper vacuum arc discharges via particle-in-cell simulations, Physics of Plasmas to be published





Comparison to experiment

self-similarity:

Crater depth to width ratio

remains constant over several orders of magnitude, and is the same for experiment and







H. Timko, F. Djurabekova et al., Mechanism of surface modification from the arc plasma-surface interaction in Cu, Phys. Rev. B 81, 184109 (2010).



Summary

- We develop a multiscale model, which comprises the different physical processes (nature and time wise) probable right before, during and after an electrical breakdown event:
 - > All the parts of the general model are pursued in parallel. We develop intense activities to cover all possible aspects.
- so Our modeling shows:
 - > Plasma is fed from the tips grown under the high electric field
 - > Tip growth can be explained by the relaxation of stresses inside of a material by the dislocation motion
 - > A dislocation-mediated mechanism can explain the high slopes of breakdown rates against the accelerating fields



RRENT PROJECTS

N COLLISON

ONO COLLISION

A GLUONE DECECTION COLLISION





ADVANCED PARTICAL COLLIDER